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Direct Evidence for Solid-like Hydrogen in a Nanoporous Carbon Hydrogen Storage Material at Supercritical Temperatures

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ABSTRACT

Here we report direct physical evidence that confinement of molecular hydrogen (H₂) in an optimized nanoporous carbon results in accumulation of hydrogen with characteristics commensurate with solid H₂ at temperatures up to 67 K above the liquid—vapor critical temperature of bulk H₂. This extreme densification is attributed to confinement of H₂ molecules in the optimally sized micropores, and occurs at pressures as low as 0.02 MPa. The quantities of contained, solid-like H₂ increased with pressure and were directly evaluated using in situ inelastic neutron scattering and confirmed by analysis of gas sorption isotherms. The demonstration of the existence of solid-like H₂ challenges the existing assumption that supercritical hydrogen confined in micropores has an upper limit of liquid H₂ density. Thus, this insight offers opportunities for the development of more accurate models for the evaluation and design of nanoporous materials for high capacity adsorptive hydrogen storage.

KEYWORDS: nanoporous materials · hydrogen storage · carbon · neutron scattering

Molecular hydrogen (H₂) has excellent potential as a sustainable, low-carbon and nonpolluting energy vector. However, above its bulk liquid—vapor critical temperature of 33 K,¹ hydrogen exists as a gas, and will not form a higher-density bulk liquid or a solid, except under extreme conditions of high pressure (e.g., > 5 GPa).²,³ Subsequently, the efficient and economic storage of molecular H₂ remains a major technological challenge.⁴,⁵ One option for increasing storage densities is via adsorption of H₂ into microporous materials, that is, materials with pore diameters <2 nm.⁶ In such materials, densification of H₂ is promoted via the enhancement of the attractive van der Waals interactions between adsorbed H₂ molecules and the solid substrate, arising from overlapping potentials from opposite pore walls. Evaluation of gas storage capacities of promising nanoporous materials generally involves measurement of the Gibbs excess uptake via isothermal gas sorption, with the subsequent conversion of the excess to absolute H₂ uptake requiring an estimate of the adsorbed hydrogen density.⁷ As the density of H₂ inside the micropores is difficult to probe experimentally, the

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maximum (limiting) density of $H_2$ is generally approximated to be the same as liquid hydrogen (i.e., a uniform density of $\sim 77$ kg m$^{-3}$ at the triple point), despite temperatures of adsorptive storage typically exceeding the bulk critical temperature.$^8$ One of the few experimental techniques that can directly access information on the state of the adsorbed $H_2$ inside a porous material and potentially validate the assumption of liquid-like adsorbed phase densities is neutron scattering. Neutrons are highly sensitive to $^1H$ due to its large incoherent neutron scattering cross section. While there have been numerous neutron diffraction studies investigating the binding of hydrogen to strong adsorption sites in metal–organic frameworks (MOFs),$^9$–$^{11}$ due to the magnitude of the incoherent scattering background, $^1H_2$ is almost always substituted by $D_2$. The differences in molecular weights may introduce isotope effects that will affect bond distances, vibrational energies and packing densities. To avoid the need for deuteration and to allow investigation of noncrystalline materials, in this study we used inelastic neutron scattering (INS), a technique that is not hampered by the $^1H$ incoherent scattering and which measures vibrational motions and thus the binding strength of atoms and molecules.

While INS measurements are typically performed at temperatures below 25 K to maximize resolution of the vibrational spectra,$^{12,13}$ here we combined INS measurements with volumetric gas sorption experiments to probe the phase behavior of supercritical hydrogen at 77 K (a temperature that is more practically relevant for $H_2$ storage applications) in a nanoporous carbon material and show direct physical evidence for an accumulation of solid-like $H_2$ in the pores.

The INS measurements were only possible due to modifications to the TOSCA instrument at the ISIS neutron facility, which enabled measurement at high resolution ($\Delta E/E < 1.25\%$, where $E$ is the energy lost by the incoming neutron) over the widest range of energy transfer of any INS instrument in the world. The improved high resolution at low energies allowed quantitative analysis of the elastic region of a scattering spectrum (where little or no energy is transferred between the incident neutron and the target $H_2$ molecule) as a function of gas pressure, with simultaneous monitoring of the inelastic regions, to provide information on the state (gaseous, liquid or solid) of the $H_2$ in the pores. INS spectra were collected on $H_2$ dosed onto a standard reference material of TE7 activated carbon beads (from MAST Carbon International, UK) at eight gas pressures (0.016, 0.070, 0.160, 0.301, 0.630, 0.998, 2.070, and 3.500 MPa), with $\sim 12$ h data collection periods. This material was selected as it presents a reasonably chemically homogeneous adsorbing surface, allowing unambiguous analysis of INS spectra, and has a modal nanopore diameter within the pore size range for maximizing interactions with $H_2$.$^{14}$

**RESULTS AND DISCUSSION**

The INS energy loss spectra of the $H_2$ in the pores at 77 K (Figure 1b) showed two important and remarkable features at each $H_2$ loading pressure measured. First, the intense, sharp peak at $\sim 0$ meV due to elastic scattering by ortho-$H_2$ indicated the presence of dense liquid- or solid-like hydrogen. While both condensed phases will show a sharp elastic scattering peak due to far higher densities of ortho-$H_2$ molecules than in the gas phase, the elastic peak from the more mobile liquid phase will typically be broadened relative to the same

![Figure 1](https://example.com/figure1.png)

**Figure 1.** (a) The para-$J (0)$ and ortho-$J (1)$ spin isomers of hydrogen. (b) INS spectra for $H_2$ adsorbed on TE7 carbon beads at 77 K in order of ascending $H_2$ pressure (0.016–3.5 MPa). (c) Magnified elastic region (1–2 meV) of the INS spectrum at the lowest pressure (0.016 MPa); the bar represents the instrumental fwhm resolution. (d) Magnified region of the INS spectra showing the rotor line at 14.7 meV, plotted on a logarithmic scale on the x-axis. ($\sim$0.7 nm as determined via $N_2$ sorption at 77 K; see Figure S1 in Supporting Information). The chemical and structural properties of the TE7 carbon were thoroughly characterized$^{15}$ and the excess $H_2$ sorption isotherm for this material was measured at 77 K to a maximum pressure of 17 MPa (see Figure S2 in Supporting Information). For the INS experiment, a room-temperature equilibrium mixture of para-$H_2$ ($J = 0$, where $J$ is the rotational quantum number) and ortho-$H_2$ ($J = 1$) was used for the in situ gas dosing at 77 K (see Figure 1a). The use of so-called “normal hydrogen”, which includes the para and ortho nuclear spin isomers of molecular hydrogen resulted in distinct characteristic features in both the elastic and inelastic regions of the neutron scattering spectrum. Due to the paramagnetic nature of activated carbons, the statistical population of para-$H_2$:ortho-$H_2$ at 77 K rapidly equilibrated to a 1:1 mixture in the sample.$^{16}$
peak from solid H₂, as a result of quasielastic interactions. The full-width at half-maximum (fwhm) of the elastic peak of \( \sim 0.3 \text{ meV} \) at the lowest adsorption pressure measured here (0.016 MPa H₂) remained approximately equal to TOSCA’s instrumental peak resolution (indicated by the horizontal bar of width 0.3 meV in Figure 1c), taking into account the significant contributions from multiple-scattering and self-shielding from the >10 cm⁻³ sample size, suggesting that the H₂ contributing to this peak had limited mobility. A full analysis of the fwhm of the elastic line is provided in the Supporting Information (see Figures S3 and S4). The second prominent feature present in all of the spectra was a well-resolved peak at \( \sim 14.7 \text{ meV} \) (Figure 1d), which is only present for the \( \text{para-to-ortho} \) transition in immobilized H₂.¹²,¹³,¹⁷,¹⁸ This peak (commonly referred to as the “rotor line”, as it corresponds to the free, unperturbed rotation of molecular H₂) corresponds solely to the amount of solid-like H₂ present. Standard deviations from the superior count statistics) in arbitrary units was scaled to result in a least-squares best fit to the calculated weight percent of H₂ in the adsorbed phase from volumetric gas sorption measurements. The original and scaled data are in Tables S1 and S2 in Supporting Information, along with details of the experimental procedure and apparatus at 77 K (Figure 3). The integrated intensity over the elastic region of the spectrum (chosen for comparison due to the superior count statistics) in arbitrary units was scaled to result in a least-squares best fit to the calculated weight percent of H₂ in the adsorbed phase from volumetric gas sorption measurements. The scaled INS intensities were found to be strongly correlated to the calculated total amount of adsorbed H₂ expressed in weight percent relative to the dry, evacuated carbon sample (see Figure 3 and Tables S1 and S2 in Supporting Information), with the concurrence of the onset of the plateau region in the INS intensities and the gas sorption data indicating that the accumulation of the solid hydrogen has an effective upper limit, which may signify the point at which the nanopores are completely filled with adsorbed H₂. The original and scaled data are in Tables S1 and S2 in Supporting Information, along with details of the modeling.

Figure 2. (a) The high pressure cryogenic cell used in the INS experiments. (b) SEM image of the morphology of the TE7 nanoporous carbon beads. (c) The total inelastic signal (integrated intensity from 2 to 500 meV). (d) Integrated intensity under the elastic peak from \(-2 \text{ meV} \) to \(+2 \text{ meV} \). (e) Integrated intensity under the 14.7 meV rotor line which indicates the amount of solid-like H₂ present. Standard errors from the fit are shown, with standard errors for (c) and (d) within the size of the data markers. The data points are joined by straight lines as guides to the eye.
Theoretical predictions of the density of the adsorbed H$_2$ (termed the “adsorbate”) in the pores, using a development of earlier analysis$^{19,20}$ applied to the experimental high-pressure Gibbs excess isotherm (to 17 MPa) for H$_2$ adsorbed on the TE7 carbon beads at 77 K, was also consistent with the evidence from INS for the formation of solid-like H$_2$. The excess data was modeled using the Tóth equation$^2$ for adsorbate filling of the pore space, to yield an estimate of the density of the adsorbed H$_2$ phase (assumed constant) of 101 ± 2 kg m$^{-3}$ (see Figure S2 in Supporting Information). This is significantly higher than the maximum density of liquid H$_2$, 77 kg m$^{-3}$ at the liquid—solid—vapor triple point (13.96 K and 0.00736 MPa)$^1$ and is closer to the density of bulk solid H$_2$, which is a highly compressible solid with densities >87 kg m$^{-3}$.\textsuperscript{16}

The powerful combination of INS measurements and gas sorption analysis points to a bulk densification phenomenon that, while consistent with modeling (see, for example, the prediction of a “solid-like phase with densities higher than bulk solid hydrogen” from Dundar et al.’s modeling of supercritical hydrogen sorption on MOFs),\textsuperscript{22} had not been previously observed experimentally. Past experimental observations consistent with localized regions of solid-like densities of adsorbed H$_2$ at elevated temperatures have generally been attributed to the strength of specific adsorption sites in MOFs and zeolites. For example, refinements of neutron diffraction data at 4 K indicate that some crystalline MOF materials support D$_2$–D$_2$ intermolecular separation distances that are shorter than the 0.36 nm found in solid H$_2$.\textsuperscript{10,23–25} In these studies, the local surface densification of H$_2$ was ascribed to the strong interactions between the H$_2$ and the unsaturated metal centers in the frameworks. Similarly, H$_2$ rotor lines previously reported in INS spectra of zeolite samples at temperatures up to 70 K show a shift in energy, due to the influence of the strong binding of H$_2$ on specific adsorption sites.\textsuperscript{18,26} Short range order and liquid-like behavior of D$_2$ has also been predicted to exist in areas between metal sites in zeolites and MOFs at 50 K, to explain broadening effects in neutron diffraction patterns.\textsuperscript{11} Carbon surfaces, however, are known to only have very weak interactions with H$_2$. The phenolic resin-derived TE7 carbons, in particular, have been shown via temperature-programmed desorption to have only small proportions of surface oxygen groups,\textsuperscript{27,28} meaning that they are likely to have limited surface functionality. This indicates that the pseudocondensation of supercritical gas seen here at 77 K is instead due to confinement effects in optimally sized nanopores.\textsuperscript{14}

Comparative, single pressure (1 bar) H$_2$ dosing INS measurements on onion-like carbon nanomaterials (OLC-1750) having negligible proportions of nanopores less than 10 Å in diameter\textsuperscript{29,30} show no such peak in the 14.7 meV region, supporting the hypothesis that the pore dimension is a critical factor. Similarly, while other activated carbon materials (TE7–20 and TE3 from MAST carbon) and AX-21 (Anderson Development Co) show small rotor peak contributions (see Supporting Information Figure S9), the integrated intensity is not proportional to the total micropore volume, indicating that only a fraction of the micropores contribute to this effect and that a very narrow pore size distribution is required. This has been shown experimentally in the case of carbide-derived carbons,\textsuperscript{31} for which higher than liquid H$_2$ densities were calculated from 77 K sorption isotherms for ~0.6 nm diameter pores.\textsuperscript{32}

Molecular simulation and modeling of hydrogen in carbon nanomaterials seems to support the possibility of densification of adsorbed hydrogen to greater than liquid densities. For example, there have been estimates of adsorptive capacities from Grand Canonical Monte Carlo molecular simulations (with quantum effects estimated using the Feynman-Hibbs effective potential) and density functional theory (without the dispersion correction) that predicted elevated levels of densification for H$_2$ in 0.3 nm carbon slit pores and carbon nanotubes,\textsuperscript{33,34} equating to densities in the region of 80 kg m$^{-3}$ at 0.1 MPa and 77 K, while theoretical modeling of supercritical sorption isotherms has predicted a transition to a solid-like phase of H$_2$ in activated carbon at 40 K.\textsuperscript{35} Experimental room-temperature small-angle neutron studies of 0.9 nm pores in carbon have also estimated greater than liquid densities of H$_2$ at pressures of 20 MPa.\textsuperscript{36}

Confinement effects are known to induce shifts in the phase transition temperatures for subcritical adsorbed phases\textsuperscript{37} and, thus, the pseudocondensation of supercritical gas reported here could be a general phenomenon. It is, therefore, possible that the presence of the ~14.7 meV rotor line in INS studies of other highly nanoporous carbons at supercritical
temperatures is likely to be similarly indicative of the presence of densified solid-like H₂. (Supporting Information, Figure S9). This is, however, the first time that the observation of this phenomenon has been correlated to gas sorption measurements and shown to be consistent with accumulation of solid-like H₂ with pressure.

CONCLUSION

Experimental evidence for a solid-like H₂ in the pores of TE7 carbon at 77 K clearly demonstrates the potential for further development of adsorptive hydrogen storage materials containing micropores of an optimum size. Due to its relatively low micropore volume (<0.5 cm³ g⁻¹, see Supporting Information) the 3 wt % H₂ uptake of the TE7 material at 77 K and 17 MPa (which is ~80% of the estimated maximum uptake at 700 µm h⁻¹, up to three spectra being collected at each pressure over collection periods of 8–12 h (pressures = 0.016, 0.070, 0.160, 0.301, 0.630, 0.998, 2.070, and 3.500 MPa). The data processing and peak integration was performed using the Mantid software (available from http://www.mantidproject.org).

METHODS

The reference sample of TE7 activated carbon beads (sourced from MAST Carbon International, UK) was produced from a carbonized phenolic resin-based material activated at high temperature (900 ºC) in a carbon dioxide atmosphere. The BET nitrogen specific surface area at 77 K was measured to be 960 ± 50 m² g⁻¹ obtained from low pressure (up to 0.1 MPa) nitrogen sorption measurements at 77 K with a 60 min equilibration time. The micropore volume, evaluated from the Dubinin–Radushkevich method is 0.43 ± 0.03 cm³ g⁻¹. The skeletal density of the sample was measured using a He pycnometer (Micromeritics AccuPyc1330) and was established as being 1.90 ± 0.03 g cm⁻³. The OLC-1750 carbon onions where synthesized by vacuum annealing of detonation nanodiamond powder (Ray Technologies Ltd, Israel). Placed in a graphite crucible, the nanodiamond powder was annealed in vacuum (10⁻⁵ to 10⁻⁷ mbar) for 3 h at 1750 ºC in a water-cooled high temperature furnace with tungsten heaters (Thermal Technology, USA).

High-pressure (up to 20 MPa) hydrogen adsorption/desorption measurements were performed on a Hiden Isochema HTP-1 Sieverts-type volumetric gas sorption analyzer with ultra-high purity (Air Products BIP-Plus, 99.9999% purity) hydrogen at 77 K using a liquid nitrogen bath for temperature control. Prior to hydrogen uptake measurements the ~150 mg samples were degassed at 623 K for 8 h under a vacuum of 0.1 MPa prior to each isothermal measurement in order to remove moisture and desorbed gases from the surface. All isotherms were fully reversible and repeat isotherms for different samples were reproducible to within 0.3% of measured amounts adsorbed.

The INS spectra were collected on the TOSCA inelastic neutron scattering beamline at the Rutherford Appleton laboratories in the UK, which has an energy window from ~3 meV to ~500 meV. The full-width at half-maximum instrumental resolution is 300 µeV over the range ~3 meV to ~3 meV (i.e., in the elastic region) and in the range 3 to 500 meV the energy resolution is ΔE/ΔE < 1.25%. A ~10 g sample of carbon was degassed via heating ex-situ at 623 K for 8 h under high vacuum (0.1 mPa), then loaded into an Air glovebox into a high pressure (7 MPa) stainless steel sample can. Temperature control was supplied by a standard cryofurnace ancillary. Normal hydrogen gas (Air Liquide, 99.999% purity) was dosed into the sample and thermally equilibrated at 77 K before the pressure was recorded using a baratron and a high-pressure transducer. The data were corrected for the presence of terminal H atoms in the carbon by subtraction of 12 h background scans of the degassed sample under dynamic vacuum at the measurement temperature (77 K). Note that the spectra for the degassed carbon sample did not show a peak at ~14.7 meV. Data were accumulated for 700 µm h⁻¹, with up to three spectra being collected at each pressure over collection periods of 8–12 h (pressures = 0.016, 0.070, 0.160, 0.301, 0.630, 0.998, 2.070, and 3.500 MPa). The data processing and peak integration was performed using the Mantid software (available from http://www.mantidproject.org).

REFERENCES AND NOTES
